# First-Principles $G_0W_0+BSE$ Calculations: Electronic and Optical Properties of Zns Monolayer

Ibrahim Bashir Usman, Auwalu Musa\*, Abdullahi Lawal, Aliyu Muhammad Received: 02 July 2025/Accepted: 27 August 2025/Published online: 05 September 2025

We investigated the electronic Abstract: structure and optical properties of 2D graphene-like wurtzite ZnS using firstprinciples calculations and many-body perturbation theory (MBPT). The structural properties were determined through firstprinciples calculations based on density functional theory (DFT). Structural optimization yields lattice constants of 3.81 Å for bulk and 3.76 Å for the monolayer, with Zn-S bond lengths of 2.33 Å and 2.21 Å, respectively and these values are in good agreement with experimental results. The quasiparticle band structure, excitonic, and optical properties were computed using manybody perturbation theory (MBPT) within oneshot  $GW(G_0W_0)$  approximation and the Bethe-Salpeter equation (BSE) approach, specifically  $G_0W_0$ -BSE. The electronic properties results show that ZnS sheet exhibits a direct band gap at the  $\Gamma$ -point, which remains unchanged as a direct semiconductor when electron-electron interactions are considered. The  $G_0W_0$ calculations verify that monolayer ZnS is a direct bandgap material with a bandgap value of 4.106 eV, which is consistent with experimental findings. The results of optical properties with inclusion of electron-hole interactions revealed that monolayer ZnS has exciton energy of 3.66 eV with binding energy of ~0.95 eV. The strong excitonic effects in the ZnS monolayer sheet make it promising for optoelectronic device applications.

**Keywords:** ZnS, monolayer, DFT,  $G_0W_0$ ,  $G_0W_0+BSE$ , Optoelectronic

## **Ibrahim Bashir Usman**

Address: Department of Physics, Bayero University, Kano, Kano State, Nigeria Email: <a href="mailto:ibrahimbashir26@yahoo.com">ibrahimbashir26@yahoo.com</a>

Orcid id: https://orcid.org/0009-0006-9681-1853

#### Auwalu Musa

Address: Department of Physics, Bayero University, Kano, Kano State, Nigeria Email:amusa.phy@buk.edu.ng

#### **Abdullahi Lawal**

Department of Physics, Ahmadu Bello University, Zaria, Kaduna State, Nigeria Email: abdullahikubau@yahoo.com Orcid id: https://orcid.org/0000-0003-1294-3180

#### Aliyu Muhammad

Department of Physics, Ahmadu Bello University, Zaria, Kaduna State, Nigeria **Email:** aliyumuhammad93@gmail.com

#### 1.0 Introduction

Research on semiconductor materials with different crystalline phases has been a key driver of the development of modern electronic and optoelectronic applications. Among these materials, sulfide zinc (ZnS)exhibits polymorphism, mainly existing in the cubic zinc blende (zb-ZnS) and hexagonal wurtzite (wz-ZnS) structures (Shahrokhi, 2016), each distinctive demonstrating structural electronic properties. The wurtzite phase of ZnS, which is characterized by hexagonal lattice structure, presents a slight but critical difference in atomic coordination and lattice structure compared to its cubic counterpart, which has a major influence on the electronic and optical properties of the ZnS molecule. The most important application of ZnS is in lightemitting diodes, lasers, flat-panel displays, and photocatalysts (Majidiyan Sarmazdeh et al., 2017). ZnS is a semiconductor with a direct bandgap (Eg = 3.72 eV in the cubic zinc blend phase and Eg = 3.77 eV in the hexagonal wurtzite phase) that promises a number of novel applications (Sharma et al., 2019). The choice of structure, size and dimension (bulk versus monolayer) has a significant influence on its properties. In recent years, the 2D monolayer form of ZnS has been attracting increasing interest because of its potential for novel electronic and optical applications. Although the monolayer of ZnS has not been synthesized yet, thin film ZnS with thickness of 11 Å in wurtzite phase have been synthesized (Majidiyan Sarmazdeh et al., 2017). The electronic transition between the valence band (VB) and conduction band (CB) produces optical properties, like photoluminescence (PL) emission and absorption. The reduced dimensionality of the material significantly impact on the electronic structure of these bands (Kumbhakar et al., 2021). Also, the interaction of materials with optical waves and photons is strongly dependent on the structure, which can then be used to control light field distribution and light propagation (Flory, 2011). Similarly, change in dimension show enhanced electronic and optoelectronic properties. Density Functional Theory (DFT) provides a computational approach investigate these properties by allowing for the calculation of the structural, electronic properties as well as optical spectrum based on atomic structure. These theoretical insights provide the basis for interpreting experimental data and customizing the properties of materials for optoelectronic applications. within DFT. the Generalized Gradient Approximation (GGA), especially the Perdew-Burke-Ernzerhof (PBE) functional (John P. Perdew, Kieron Burke, 1966), provides a balance between computational efficiency and reasonable accuracy. However, a known limitation of GGA-based functionals is the

systematic underestimation of semiconductor band gaps. This shortfall arises primarily from the self-interaction error and the insufficient treatment of exchange–correlation interactions (John P. Perdew, Kieron Burke, 1966), (Becke, 1993).

Sharma et al., (2019) analyzed wz-ZnS and zb-ZnS using GGA, The reported band gaps was 2.06eV and 2.01eV, for wz-ZnS and zb-ZnS respectively, indicating underestimation of around 45% comparing to the experimental values of 3.72 eV for zb and 3.77 eV for wz-ZnS (Ves et al., 1990; Karazhanov et al., 2006; Voigt et al., 2020) reported that the GGA-calculated band gap for bulk ZnS was 2.03eVand this value is smaller than the experimental value. Conversely, the application of GGA+U adjusted the band gap to 3.57eV, illustrating the impact of adding onsite Coulomb interaction corrections. Similarly, in monolayer forms of ZnS, GGA functionals often yield band gaps in the range of 2.27-2.65eV (Lashgari et al., 2016), (Shahrokhi, 2016). These values are improved adopting functionals such as  $G_0W_0$ approximations, which can correct the band values more gap to consistent experimental data. These investigations highlight in general the shortcomings of standard DFT-GGA in accurately estimating the band gap of ZnS and the effectiveness of using corrections for achieving results more corresponding to experimental observations. Understanding and addressing these inconsistencies are critical for the design and application of ZnS-based materials in optoelectronic devices.

For decades, researchers have used standard density functional theory (DFT) and the GW approximation to study the optical properties of materials (Gilmore et al., 2015; Lawal et al., 2021). However, these methods have limitations, particularly in describing excitonic effects and electron-hole interactions, which are crucial for accurate optical spectra predictions. The mentioned approaches often





yield significant discrepancies with experimental results (Li et al., 2024; Yusuf et al., 2024). To overcome this, a two-particle approach by solving Bethe-Salpeter equation (BSE) based on GW corrections can be employ. This method provides a more accurate description of optical properties, capturing the influence of electron-hole interactions and yielding results in better agreement with experimental data.

This study presents calculations of the structural properties of ZnS in monolayer using various exchange-correlation potentials within the framework of density functional theory (DFT) as implemented in Quantum Espresso. Additionally, electronic and optical properties are investigated using many-body perturbation theory (MBPT) via one-shot  $G_0W_0$  approach and the solution of the Bethe-Salpeter equation (BSE), as implemented in the YAMBO al., package (Marini et 2009). understanding of the electronic and optical properties of ZnS monolayers using G0W0+BSE will provide crucial theoretical designing next-generation insights for optoelectronic devices and guide future experimental synthesis of 2D ZnS materials

#### 2.0 Computational Methods

The calculations were performed using the first principle methods based on DFT with plane wave as the basis set as implemented in the Quantum ESPRESSO (Open Source package for research in electronic structure, simulation and optimization, QE) simulation package (Giannozzi et al., 2009) and Many Body Perturbation Theory within GW approximation as implemented in YAMBO code (Marini et al., 2009). In the DFT calculation, exchangecorrelation functional was approximated by the Gradient Approximation Generalized Perdew-Burke-Ernzerhof (GGA-PBE) (John P. Perdew, Kieron Burke, 1966), to treat electron-electron interaction. First, the initial structure of bulk wurzite-structured Zinc Sulfide (wz-ZnS) was obtained from materials

project of (Jain et al., 2013) and optimized using variable-cell relaxation (vc-relax), using GGA-PBE method. Then, a  $7\times7\times1$  k-points for the integration of the first Brillouin zone was chosen, while denser values of 12×12×1 were used for the density of states using a technique known as Marzarri-Venderbilt smearing. Plane waves with kinetic energy cutoffs of 50Ry to expand the electronic wavefunctions and 320Ry for charge density was used. Five other exchange correlations functional (BP, WC, PZ, PBE-sol and revPBE) were used to calculate the band structure. A vacuum layer of 20 Å was used in perpendicular direction (c-axis) of ZnS monolayer along the z-direction to avoid inter layer interaction. However, to correct the electronic band structure obtained from standard DFT approach, we performed GW calculations within  $G_0W_0$  approximation as implemented in YAMBO package (Marini et al., 2009) to obtain a real quasiparticle (QP) energies (Eq.1) correction to the Kohn-Sham (KS) eigenvalues  $E_{nk}^{DFT}$  (Lucarini et al., 2005).  $E_{nk}^{QP} = Z_{nk} \langle \varphi_{nk}^{DFT} | \Sigma_{GW}(E_{nk}^{DFT}) - V_{XC} | \varphi_{nk}^{DFT} \rangle +$ 

where  $V_{XC}$  is the DFT exchange-correlation potentials,  $\Sigma_{GW}$  is the GW self-energy which is the product of one-particle Green's function, G and screened Coulomb potential, and k and n are the k-point and band indices respectively,  $E_{nk}^{DFT}$  and  $\varphi_{nk}^{DFT}$  are the KS eigenvalues and eigenfunctions respectively, and  $Z_{nk}$  is the orbital renormalization factor.

For calculating excitonic and optical properties using YAMBO, the Kohn-Sham (KS) energies and wave-functions from DFT calculations serve as input for the Bethe-Salpeter equation (BSE) calculations. The BSE is then solved using the Tamm-Dancoff approximation and interaction kernel, which accounts for dynamic screening and electron-hole interactions. This yields electron-hole interaction and excitation energies. Finally, the optical absorption spectrum is calculated, including the effects of electron-hole interactions, providing a more





accurate representation of the material's optical properties. On the other hand, for optical properties calculations, the dielectric function,  $\varepsilon(\omega) = \varepsilon_1(\omega) + i\varepsilon_2(\omega)$ , is investigated in terms of incident photon energy using random phase approximation (RPA). The real and imaginary parts of the dielectric function are denoted by  $\varepsilon_1(\omega)$  and  $\varepsilon_2(\omega)$ , respectively (Lawal et al., 2017a), (Arbi et al., 2012).

$$\varepsilon_{1}(\omega) = 1 + \frac{2P}{\pi} \int_{0}^{\infty} \frac{\omega' \varepsilon_{2}(\omega')}{\omega'^{2} - \omega^{2}} d\omega' \qquad (2)$$

$$\varepsilon_{2}(\omega) = \frac{16\pi e^{2}}{\omega^{2}} \sum_{S} |\vec{\lambda} \langle O|\vec{v}|S \rangle|^{2} \delta(\omega - \Omega^{S})$$
(3)

where  $\vec{\lambda}$  is the polarization vector of light  $\langle O|\vec{v}|S\rangle$  is the optical transition matrix from valence to conduction states and is the principal value of the integral and the integral is over irreducible Brillouin zone. Understanding both the real and imaginary components of the dielectric function enables the calculation of optical properties. We calculated the refractive index  $n(\omega)$ , the extinction coefficient  $k(\omega)$ , the absorption coefficient  $\alpha(\omega)$ , the reflectivity  $R(\omega)$ , loss function  $L(\omega)$ , and the conductivity  $\sigma(\omega)$  using the following equations (Arbi et al., 2012):

$$n(\omega) = \sqrt{\left(\frac{\sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega) + \varepsilon_1(\omega)}}{2}\right)}$$
 (4)

$$k(\omega) = \sqrt{\frac{\sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega) - \varepsilon_1(\omega)}}{2}}$$
 (5)

$$\alpha(\omega) = \frac{\omega}{c} \sqrt{2\left(\sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)} - \varepsilon_1(\omega)\right)}$$
(6)

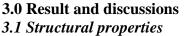
$$R(\omega) = \left| \frac{\sqrt{\varepsilon(\omega)} - 1}{\sqrt{\varepsilon(\omega)} + 1} \right|^{2}$$

$$L(\omega) = \frac{\varepsilon_{2}(\omega)}{\varepsilon_{1}^{2}(\omega) + \varepsilon_{2}^{2}(\omega)}$$

$$\sigma(\omega) = \frac{\omega \varepsilon_{2}(\omega)}{4\pi}$$
(9)

$$L(\omega) = \frac{\varepsilon_2(\omega)}{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)}$$
 (8)

$$\sigma(\omega) = \frac{\omega \varepsilon_2(\omega)}{4\pi} \tag{9}$$



The crystal structures of bulk and monolayer ZnS were optimized using density functional implemented theory as in Quantum ESPRESSO. Geometry relaxation performed until the atomic forces were reduced below  $1\times10^{-3}$  Ry/Bohr, and the total energy difference between iterations was less than  $1\times10^{-6}$  Ry. For the bulk zinc wurtzite ZnS, the calculated equilibrium lattice constant was 3.81 Å, which compares well with the reported experimental value of about 3.82 Å (Kushwah et al., 2020) and theoretical work (Lashgari et al., 2016). The corresponding Zn-S bond length was obtained as 2.33 Å. In the case of the ZnS monolayer, a vacuum gap of about 20 Å was added along the out-of-plane direction to prevent spurious interactions between periodic images. After structural relaxation, the in-plane lattice constant was optimized to 3.76Å, with a Zn-S bond length of 2.21 Å. Compared with the bulk phase, the monolayer shows slightly shorter bond lengths, which can be attributed to reduced atomic coordination and stronger in-plane bonding. These findings are consistent with earlier theoretical reports (Behera & Mukhopadhyay, 2014), (Majidiyan Sarmazdeh et al., 2017). Fig. 1 shows the structures of bulk and monolayer ZnS, the comparison between bulk and monolayer ZnS reveals that reducing the dimensionality results in a modest contraction of both the lattice constant and Zn-S bond length. These structural variations are expected to influence the electronic band gap and optical absorption edge, which will be analyzed in the subsequent sections.

# 3.2 Electronic properties

To investigate electronic properties of Wz-ZnS, the band structure, total and partial density of states (DOS) are calculated. Fig. 3 illustrate the band structures along the high symmetry  $\Gamma$ -Mdirections using different exchange correlation functionals and G<sub>0</sub>W<sub>0</sub> approaches,





the electronic band structures were computed within six different exchange correlation approximations based on DFT in the energy range of -5 to 6eV setting Fermi energy level scale at 0eV represented by a red dash.

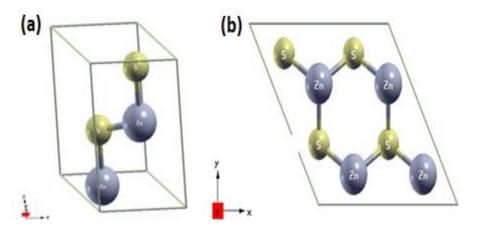


Fig 1 Structure of ZnS: (a) Bul (b) Monolayer

The obtained results are compared in table 1. Both the valence band maximum (VBM) and the conduction band minimum (CBM) for all funtionals occur at the  $\Gamma$ -point, resulting in a direct band gap. All the band gaps values obtained with the six exchange correlations are significantly smaller than experimental value. This discrepancy arises from the limitations of the DFT approach due to the approximations describing exchange-correlation used in functional. The discrepancy between DFTpredicted band gaps and experimental values can be addressed by incorporating many-body theory perturbation (GW) into calculations. This approach utilizes the Green's function to accurately capture the electronic density of the fully interacting system, leading to improved band gap predictions. accurately predict quasiparticle energies and band gaps, we applied self-energy corrections using the one-shot  $G_0W_0$  approximation within many-body perturbation theory (MBPT). The resulting G<sub>0</sub>W<sub>0</sub> band structure is shown in Fig. 2(g), providing a more reliable result of the material's electronic properties. As shown in Fig. 2(b), incorporating self-energy corrections via MBPT leads to a moderate band gap increase, resulting in better agreement with experimental values. The calculated direct

quasiparticle band gap of monolayer ZnS is 4.106~eV. Our calculations reveal that the  $G_0W_0$  approximation on top of standard DFT, accurately predicts the magnitude of the monolayer ZnS band gap. A comparison of our results with previous works are presented in Table 2.

## 3.2.1 Density of states

To clearly understand the nature of the bands in monolayer ZnS. The total (TDOS) and projected density of states (PDOS) were calculated. The TDOS and PDOS (for rev-PBE functional only) are presented in Figure 3 (a) and (b) respectively, along with the Fermi energy level scale at 0 eV shown in a red dash. The main contribution of lowest valence band (the energy range between -14 and -6eV) is related to the Zn-d orbital, the main contribution of states in the maximum occupied valence bands (the energy range between -7eV to Fermi level) is related to S atom p-orbital and contribution of states in the conduction region (above 1 eV) is related to Zn-s orbital. Thus, for monolayer ZnS, the S atom p-orbital has the highest contribution in valence bands, while Zn atom s-orbital has the highest contribution in conduction bands.





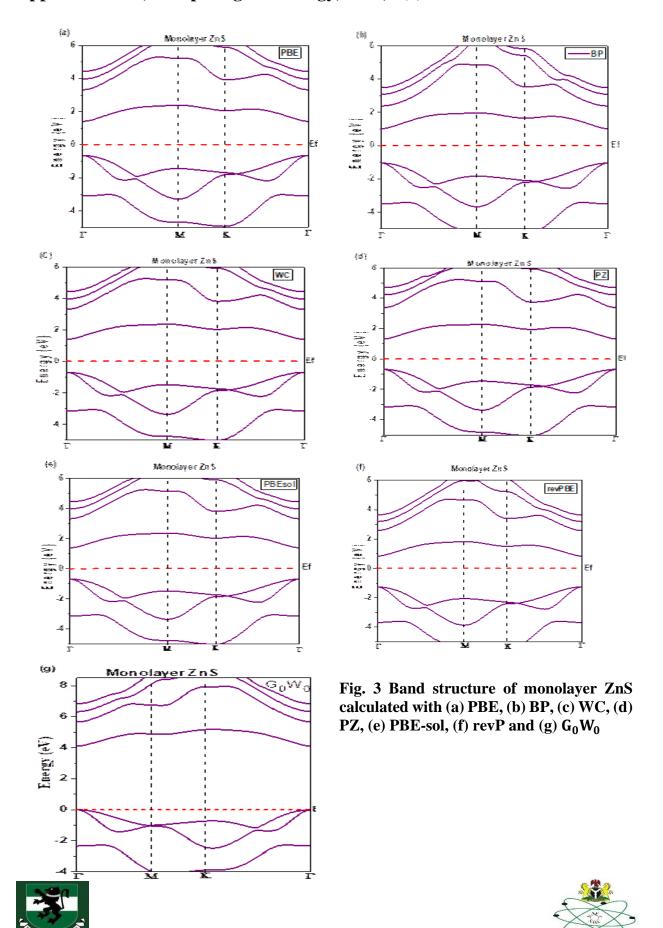
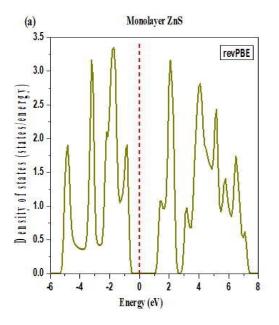


Table 2: Comparison of energy band gap (eV) values calculated with local density approximation (LDA), generalized gradient approximation (GGA) and  $G_0W_0$  results for monolayer ZnS

Methodology	Band Gap (eV)	Type of Gap
PBE	2.070	Direct
BP	2.082	Direct
WC	2.104	Direct
PZ	2.014	Direct
PBEsol	2.086	Direct
revPBE	2.112	Direct
$G_0W_0$	4.106	Direct
Previous Theoretical Work		
GGA (Shahrokhi, 2016)	2.650	Direct
GGA (Lashgari et al., 2016)	2.270	Direct
LDA (Behera & Mukhopadhyay, 2014)	2.622	Direct



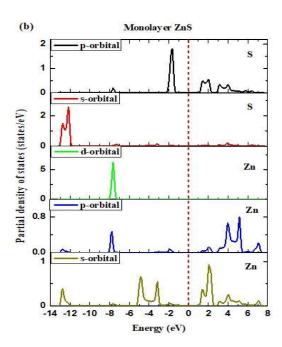


Fig. 5 (a) Total density of state and (b) Partial density of states of Monolayer ZnS.

# 3.3 Optical properties

The optical properties of ZnS were studied using a combination of **Quantum ESPRESSO** and **YAMBO**. First, ground-state electronic structure calculations were performed using Quantum ESPRESSO within the generalized gradient approximation (GGA). The resulting Kohn-Sham wavefunctions were then used as

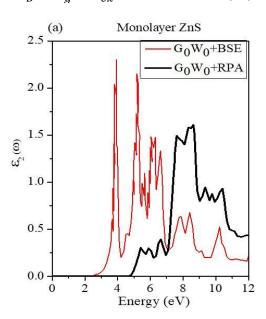
input for optical property calculations in YAMBO. The optical response of the material was first analyzed by examining its dielectric function. Fig. 4(a) and (b) represent the real and imaginary part of dielectric function of monolayer ZnS for polarization along a parallel direction to the hexagonal axis. The imaginary part of the dielectric function,  $\varepsilon_2(\omega)$ , computed





G<sub>0</sub>W<sub>0</sub>+RPA, reveals the absorption threshold around the quasiparticle gap, coupled with a broad feature signifying interband transitions. However, it does not account for excitonic effects needed dimensional materials. When excitonic effects are included by solving Bethe-Salpeter Equation (BSE) in addition to G<sub>0</sub>W<sub>0</sub>, the optical spectrum changes significantly. A strong peak of 3.66 eV appears below the quasiparticle band gap, which clearly indicates the presence of tightly bound excitons. The energy difference between this exciton peak and the band gap gives an estimated exciton binding energy of about 0.950 eV. The binding energy of the lowest energy exciton (EB) was computed by (Shahrokhi, 2016);

$$E_B = E_g - E_{ex} \tag{10}$$



where  $E_g$  is the QP band gap and  $E_{ex}$  is excitation energy. This result confirms that weak dielectric screening in 2D materials enhances excitonic interactions.

The real part of the dielectric function reflects how the material responds to an external electric field. Dielectric constant at higher frequency in the perpendicular direction  $\varepsilon_{\perp}(\infty)$  for  $(G_0W_0 + \text{BSE})$  is around 1.73; it reaches a peak at 4.14 eV near the absorption onset and decreases gradually at higher photon energies. For  $(G_0W_0 + \text{RPA})$  approach, the dielectric constant start rising at 1.63. The static dielectric constant is found to be higher when excitonic effects are included, due to stronger polarization.

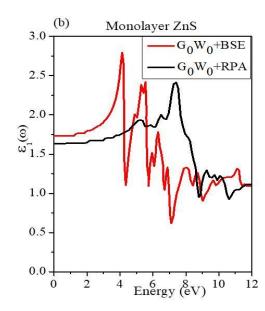


Fig. 4 (a) Imaginary part of the dielectric function (b) Real part of the dielectric function of monolayer ZnS calculated using  $G_0W_0 + \text{RPA}$  and  $G_0W_0 + \text{BSE}$ 

The optical refractive index  $n(\omega)$ , derived from the dielectric function, help describe the material's interaction with light. Fig 5(a) and (b) shows the refractive index  $n(\omega)$  and reflectivity  $R(\omega)$  of monolayer ZnS as function of energy. The static refractive index is slightly higher in the  $G_0W_0+BSE$  case, reflecting an increased ability to slow down light.  $n(\omega)$  rise

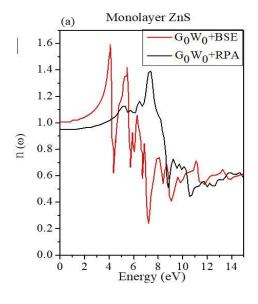
steeply at lower photon energies, attains high refractive index of 1.59 within UV region and decline at higher energies, consistent with the observed absorption behavior. The reflectivity  $R(\omega)$  spectrum reveals how much light is reflected at different photon energies. From Fig 4(b), the spectrum is relatively low in the visible and near-UV range, the first edge with  $G_0W_0+BSE$  was found to be 4.573%. The





reflectivity spectrum starts to increase from 4.573%, attaining maximum level of 30.859% corresponding to photon energy value 6.826 eV, suggesting the material is largely transparent at those energies. Also, the value

reflectivity spectrum at zero energy using  $G_0W_0$ +RPA was 1.291%. The low reflectivity across the entire spectrum makes the material suitable for applications that require high transparency or minimal reflection.



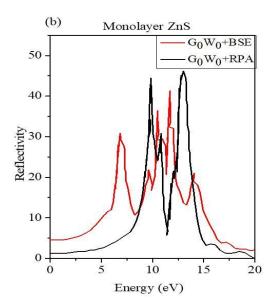
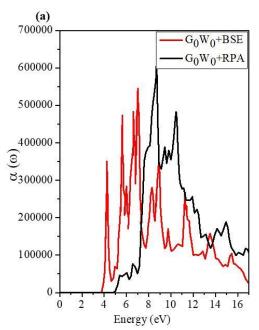


Fig. 5 (a) Refractive index (b) Reflectivity of monolayer ZnS calculated using  $G_0W_0 + \text{RPA}$  and  $G_0W_0 + \text{BSE}$ .



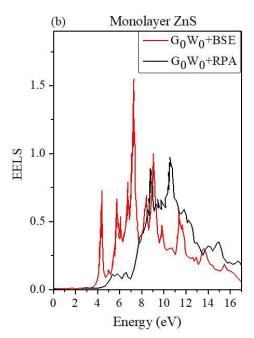


Fig. 6 (a) Absorption coefficient (b) Electron Energy Loss Spectrum (EELS) of monolayer ZnS calculated using  $G_0W_0 + \text{RPA}$  and  $G_0W_0 + \text{BSE}$ .





The absorption coefficient  $\alpha(\omega)$  provides insight into how effectively the material absorbs light across different energies. Fig 6(a) absorption coefficient displays the monolayer ZnS in two approaches. In the G<sub>0</sub>W<sub>0</sub>+RPA spectrum, absorption starts around the corrected band gap and spans a broad range into the ultraviolet. After accounting for electron-hole interactions through the BSE, a pronounced peak appears just below the absorption edge at around 3.92 eV. The absorption edge is also slightly shifted to higher energy compared to previous DFT result. The electron energy loss function  $L(\omega)$ reveals where collective excitations, plasmons, occur. With the two approaches shown in Fig. 6, a clear peak for  $G_0W_0$ +BSE appears at the energy 7.27 eV, consistent with (Shahrokhi, 2016), signaling a plasmon resonance. This feature remains mostly unaffected by the inclusion of excitonic effects, as it arises from higher-energy collective motions rather than individual electron-hole transitions.

#### 4.0 Conclusion

In conclusion, our comprehensive investigation of the electronic structure and optical properties of 2D graphene-like wurtzite ZnS using first-principles calculations and manybody perturbation theory (MBPT) has provided valuable insights into the material's potential applications. The calculated lattice constants and Zn-S bond lengths are in good agreement with experimental results, validating the accuracy of our DFT-based optimization. The quasiparticle band structure and excitonic properties computed using the G0W0-BSE approach reveal that monolayer ZnS is a direct bandgap material with a bandgap value of 4.106 eV, consistent with experimental findings. The strong excitonic effects, characterized by an exciton energy of 3.66 eV and a binding energy of ~0.95 eV, make monolayer ZnS a promising candidate for optoelectronic device applications. These results demonstrate the potential of 2D ZnS in advancing optoelectronic technologies and highlight the importance of considering manybody interactions in understanding the material's properties.

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#### **Declaration**

## **Consent for publication**

Not applicable.

## Availability of data and materials

The publisher has the right to make the data publicly available.

#### **Ethical Statement**

This research was conducted entirely through computational simulations using first-principles and many-body perturbation theory methods. No human participants or animals were involved. All authors contributed voluntarily, maintained scientific integrity, and have been appropriately acknowledged.

## **Competing interests**

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#### **Authors' Contributions**

I.B.U. conceptualized the study, carried out the DFT and MBPT calculations, and prepared the initial manuscript draft. A.M. contributed to data analysis, interpretation of electronic and optical properties, and manuscript revision. A.L. assisted in theoretical validation, literature review, and proofreading. All authors discussed the results and approved the final manuscript.



